Supporting Information

Understanding the Role of Electronic Effect in CO on Pt-Sn alloy Surface via Band Structure Measurements

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Experimental details

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Clean Pt(111) surface is obtained from repeated cycle of Ar⁺ ion sputtering (1.5 keV, I _{emission current} =6 mA, P(Ar) = 3×10^{-6} Torr for 10 mins) and annealing (T=1000 K for 2 mins). The cleaning cycle was repeated until no carbon signal is observed. The surface cleanness was checked with Auger electron spectroscopy (AES). All the LEED and AES data were taken with a SPECS ErLEED 3000-D. The residual gases in a high vacuum chamber were checked with a mass spectrometer (SRS RGA 100), and the partial pressure of H₂, H₂O, CO and CO₂ gases were 5.9×10^{-10} , 4.0×10^{-11} , 2.8×10^{-11} and 9.1×10^{-12} Torr, respectively. No sign of carbon was observed from AES prior to the ARPES measurement. The measurement time for ARPES was 1 hour for a specific band, and 2 hours for full mapping. Under the current chamber pressure condition, with normal molecular sticking coefficients, more than 20 hours is needed to have 1.0 ML coverage of impurity molecules (H₂, H₂O, and CO₂) on Pt(111) at 100 K. During the measurement of ARPES, no significant changes were found.



Figure S1. AES data of Pt(111) (a) before and (b) after cleaning. (c) The partial pressure of residual gas measured with mass spectrometer in UHV chamber during ARPES measurement.

Surface arrangement study of Pt, Pt-Sn/Pt(111) surface alloys, and CO adsorbed surfaces using LEED.



Figure S2. LEED patterns of Pt(111), Pt-Sn/Pt(111) $\sqrt{3} \times \sqrt{3}$ surface alloy, and Pt-Sn/Pt(111) 2×2 surface alloy, and changes in the surface arrangement by CO adsorption: (a) Pt(111), (b) Pt-Sn/Pt(111) $\sqrt{3} \times \sqrt{3}$, (c) Pt-Sn/Pt(111) 2×2 , (d) ~0.25 ML CO/Pt, (e) ~0.5 ML CO/Pt, and (f) ~0.5 ML CO/Pt-Sn/Pt(111) 2×2 . Dotted lines indicate the Brillouin zone boundary.

Two different Pt-Sn/Pt(111) surface alloys, $\sqrt{3} \times \sqrt{3}$ and 2×2 , are prepared by Sn deposition and annealing on a Pt single crystal surface at 800 and 1,100 K, respectively. The surface arrangements of Pt(111) and the Pt-Sn/Pt(111) surface alloy are confirmed with LEED.^{1,2} CO is dosed on the sample surface using gas pipe close from the sample to keep the base pressure low. While changes in LEED patterns are observed between Pt and the Pt-Sn/Pt(111) 2 \times 2 surface allow with CO adsorption no such changes are observed with the Pt-Sn/Pt(111) $\sqrt{3} \times$ $\sqrt{3}$ surface; this is because CO molecules do not form an well-ordered arrangement on the $\sqrt{3} \times \sqrt{3}$ surface due to the increased proportion of Sn.³ CO adsorbates form ordered adlayers on Pt and Pt-Sn/Pt(111) 2×2 surface alloy. Two different adlayer configurations for CO/Pt and CO/Pt-Sn/Pt(111) 2×2 surface alloy are confirmed by comparing LEED patterns.^{4,5} At 0.25 ML coverage on Pt(111), CO sits on the top site and forms a $\sqrt{3} \times \sqrt{3}$ superstructure. At 0.5 ML, CO sits on both top and bridge sites with $c(4 \times 2)$ structure. In the case of of Pt-Sn 2 \times 2 surface alloy surface, no superstructure could be identified at 0.25 ML, consistent with a previous report³, while $2\sqrt{3} \times 2\sqrt{3}$ superstructure was observed at 0.5 ML coverage of CO without any mixed phase. The real space model for CO/Pt(111) and CO/Pt-Sn/Pt(111) 2×2 surface alloy can be found in references.^{3,5} During each measurement of ARPES, we checked the surface ordering with LEED, which exhibits the presence of single ordered phase. Previously, CO binding energies on Pt(111)⁶ and Pt₃Sn(111)⁷ bulk alloy surfaces were calculated using DFT. At low coverage (1/9 ML), the most stable adsorption site of CO on Pt(111) was the top site with 1.52 eV binding energy. At 0.5 ML coverage, 2 top and 2 bridge site CO atoms on the $c(4 \times 2)$ unit cell are most stable with 1.44 eV binding energy. On Pt₃Sn(111) bulk alloy surface, top site was the most stable site with 1.5 eV binding energy for 0.25 ML CO coverage. For 0.5 ML CO coverage, hollow site becomes more stable with 1.27 eV binding energy.

The atomic model for DFT calculations.



Figure S3. The atomic model of (a) 0.33 ML CO/Pt, (b) 0.5 ML CO/Pt, (c) Pt-Sn/Pt, and (d) CO/Pt-Sn/Pt. The gray, purple, black, and red atoms represent Pt, Sn, C, and O, respectively. The CO distribution of 0.5 ML CO/Pt-Sn/Pt are employed from reference.³ In actual comparison with the experimental data in Fig. 3, the atomic model of 0.33 ML CO/Pt is compared with experimental 0.25 ML CO/Pt system since both show a $\sqrt{3} \times \sqrt{3}$ LEED pattern (Figure S2) while our 0.25 ML model has a different structure.

Table S1. Calculated average ausorption energies of CC	О
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Material	E (eV)
0.5 CO ML/Pt	-1.38
0.58 CO ML/Pt-Sn/Pt	-0.93

ARPES simulation for Pt and Pt-Sn/Pt(111) 2 × 2 surface alloy.



Figure S4. (a) Pt, and (b) Pt-Sn/Pt(111) 2×2 surface alloy Fermi surfaces and band structures of (c) Pt, and (d) Pt-Sn/Pt(111) 2×2 surface alloy near the M point obtained with ARPES simulation.

The DFT calculations for ARPES simulation are based on a fully relativistic multiple scattering approach (Korringa–Kohn–Rostoker (KKR) method), as implemented in the SPR-KKR package.⁸ As the first step in these theoretical investigations, we performed self-consistent ground state calculations for 2D semi-infinite Pt(111) and reconstructed Pt-St/Pt(111) 2×2 surfaces. The self-consistent results served as an input for the spectroscopic investigations. ARPES calculations are performed in the framework of the fully relativistic one-step model of photoemission using experimental geometry and photon energy.⁹ Thus, the theory accounts for effects induced by light polarization, matrix-element effects, final state effects, and surface effects.

Orbital character calculations for Pt(111)



Figure S5. Orbital character calculations of Pt(111). Orbital contributions of (a) s, p, d, (b) d_{x2-y2} , (c) d_{xy} , (d) d_{yz} , (e) d_{zx} , and (f) d_{z2} orbitals are indicated by circles proportional to the contribution.

Figure S5 shows the orbital contribution of Pt(111) bands near the Fermi level. The orbital contribution is indicated by the size of the circle. The bands near the Fermi level of Pt(111) are mostly *d*-bands (Figure S3a) and the detailed orbital character of these *d*-bands are shown in Figures S3b-f. Figures S3b-f shows the d_{x2-y2} , d_{xy} , d_{yz} , d_{zx} , and d_{z2} orbital contributions, respectively. DFT calculations are performed with the Vienna ab initio simulation package (VASP) based on the frozen-core full-potential projector augmented-wave (PAW) method,^{10,11} under the generalized gradient approximation (GGA) of Perdew-Burke-Ernzerhof (PBE) as an exchange correlation functional.¹² We optimized the cell parameters of fcc Pt until the force became less than 10⁻⁹ eV/Å using a 7 × 7 × 7 Γ -centered K-points mesh. This resulted in a cell parameter of 2.81296 Å for a primitive unit cell. Based on this, we constructed a hexagonal supercell of fcc Pt featuring ABC stacking of three Pt triangular lattices. The electronic structure of the Pt supercell, which we will refer to hereafter as Pt(111), is relaxed to include spin-orbit coupling with a threshold of 10⁻³ eV.

Orbital character change of hexagonal band due to the CO adsorption.



Figure S6. Band structures of (a) Pt(111), (b) ~0.25 ML CO/Pt(111), and (c) ~0.5 ML CO/Pt(111) along the Γ K direction, and DFT calculation of (d) Pt(111), (e) ~0.33 ML CO/Pt(111), and (f) ~0.5 ML CO/Pt(111) along the Γ K direction. The hexagonal bands are marked with blue arrows.

Figure S6 shows the band structure of Pt(111) and CO adsorbed Pt surfaces along the Γ K direction and the DFT calculated band structure. Although it is not easy to recognize band splitting at the DFT calculation, the hexagonal bands become broader due to the CO adsorption. The orbital character calculation of the hexagonal band (Table S1) supports our intuition of surface-bulk splitting at CO/Pt(111). The Pt(111) hexagonal band is mostly made of in-plane orbital band. This in-plane orbitals have weak interaction between the layer; and cause surface-bulk splitting when CO is adsorbed on the surface. The out-of-plane orbital contribution, especially $d_{yz} + d_{xz}$ contribution of Pt(111) hexagonal band increases as CO adsorb. This orbital contribution change shows similar trend with the orbital contribution change of M point band (Table 1) and shows the electronic band structure change near Fermi level during CO adsorption comes from the back-donation π -bonding states. Calculation details for the band structure and orbital characters are written in the main text.

Table S2. Calculated orbital character change of hexagonal band. Calculated orbital contribution for the *hexagonal band* of Pt during CO adsorption on Figures S4a-c.

Material	d _{z2} contribution	$d_{yz} + d_{xz}$ contribution	$d_{xy} + d_{x2-y2}$ contribution
Pt	0.111	0.086	0.668
0.33 ML CO/Pt	0.159	0.114	0.612
0.5 ML CO/Pt	0.145	0.236	0.524

The second derivative of CO-induced energy band shifts of Pt and Pt-Sn/Pt(111) alloy



Figure S7. The second derivatives of CO-induced energy band shifts are shown at the M point of (a) Pt(111), (b) ~ 0.25 ML CO/Pt(111), (c) ~ 0.5 ML CO/Pt(111), (d) Pt-Sn/Pt(111) surface alloy, (e) ~ 0.25 ML CO/Pt-Sn/Pt(111) surface alloy, and (f) ~ 0.5 ML CO/Pt-Sn/Pt(111) surface alloy.

Figure S7 shows an intensity plot of the second derivative of Figure 3. The second derivative clearly shows the energy shifts of specific bands, indicated by yellow dotted lines in Figure 3.

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